# Redetermination of the Crystal Structure of $\gamma$ -In<sub>2</sub>Se<sub>3</sub> by Twin Crystal X-Ray Method

A. Pfitzner and H. D. Lutz

Anorganische Chemie I, Universität Siegen, D-57068 Siegen, Germany

Received April 4, 1996; accepted April 10, 1996

DEDICATED TO PROFESSOR ROGER BLACHNIK ON THE OCCASION OF HIS 60TH BIRTHDAY

The crystal structure of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> ( $P6_1$  or  $P6_5$ , Z = 6, a = 712.86(5), c = 1938.1(2) pm) was redetermined by X-ray structure determination methods confirming the results of Likforman *et al.* reported in 1978 (8). The crystals (hexagonal pyramids and bipyramids, and small plates) obtained by chemical transport with iodine as transporting agent were multiple twins. The collected data of two twinned crystal specimens were refined (final R1 = 0.0315 and 0.0559 for 1553 and 1602 independent reflections with  $I > 2\sigma_1$ , respectively). The structure is built up by In(1)Se<sub>5</sub> trigonal bipyramids and distorted In(2)Se<sub>4</sub> tetrahedra resulting in a distorted wurtzite-type-like arrangement.  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> is the room-temperature polymorph of indium(III) selenide contrary to some other reports in the literature, e.g., Julien *et al.* (1985) (3). © 1996 Academic Press, Inc.

## INTRODUCTION

The crystal structures of the various In<sub>2</sub>Se<sub>3</sub> polymorphs (1-6) have been not reported so far with the exception of two older papers on  $\alpha$ -,  $\beta$ -, and  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> (7, 8). After Osamura *et al.* (7),  $\alpha$ - and  $\beta$ -In<sub>2</sub>Se<sub>3</sub> crystallize in layer structures with indium in tetrahedral and octahedral holes formed by selenium, respectively. Likforman et al. (8) determined the structure of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> revealing two different indium sites, viz. tetrahedrally coordinated In<sup>3+</sup> ions and those in a trigonal bipyramidal coordination. These results, however, were only qualitatively considered to be confirmed (9). Thus, it was concluded from the densities of the polymorphs under discussion that the coordination of the In<sup>3+</sup> ions is equal for  $\alpha$ - and  $\beta$ -In<sub>2</sub>Se<sub>3</sub> but larger than in the case of the  $\gamma$  modification (5). Additional X-ray structure studies promised at that time (5) failed because of multiple twinning of the crystals obtained. Furthermore, the phase relationships of the In<sub>2</sub>Se<sub>3</sub> polymorphs are not fully established. In the older literature as well as that by Likforman et al. (2) (in contrast to Likforman et al. (8)) and by Julien et al. (3),  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> was claimed to be a high-temperature modification. In 1988, we argued for  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> as the roomtemperature polymorph (5). Full confirmation of this reverse order, however, is lacking until now. In some recent papers (6, 10) this problem was negated.

The structure data reported by Likforman *et al.* (8) give no hints on the normally observed twinning of this compounds although the reliability factor of the refinement  $(R = 0.066 \text{ for } I > 2\sigma_I)$  reveals some problems with the crystal studied. We therefore redetermined the crystal structure of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub>.

 $\gamma$ -In<sub>2</sub>Se<sub>3</sub> crystallizes in the enantiomorphous space groups  $P6_1$  or  $P6_5$  with 6 formula units in the unit cell. The specimen studied by Likforman *et al.* (8) belonged to space group  $P6_1$ . We obtained twins showing Laue symmetry 6/mmm (possible space groups  $P6_122$  or  $P6_522$  with respect to the reflection condition 00l with l = 6n) as well as crystals showing small deviations from Laue symmetry 6/mmm resulting in 6/m (possible space groups  $P6_1$  or  $P6_5$ ).

#### **EXPERIMENTAL**

Crystals of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> were grown by chemical vapor transport in closed silica tubes using iodine as the transporting agent (11–13). The transport occurs from the cold to the hot zone, i.e., from 700 to 800 K (13). The transporting rate depends strongly on the I<sub>2</sub> concentration in the tube. Using 5 mg/cm<sup>3</sup>, crystals as large as 2 mm in diameter were obtained within 2 days. The typical habit of the black (dark red in transmission) crystals were hexagonal pyramids, hexagonal bipyramids, and some amounts of small plates. The composition of one of the crystals has been determined by a microprobe (Cam Scan 44 with EDAX detector) to be 60.7 mol% Se (calc. 60 mol%) and 39.3 mol% In (40 mol%).

Various small plates were transferred to an Enraf-Nonius CAD 4 diffractometer. Graphite-monochromatized Mo $K\alpha$  radiation was used. The intensity data were corrected for Lorentz and polarization effects with NRCVAX (14). Empirical absorption corrections were performed by  $\psi$  scans. The variations in intensity throughout the data collection were less than 1%. The structure was refined by full-matrix least-squares refinement of positional and anisotropic thermal parameters, extinction coefficient, and scale factor using scattering factors for neutral atoms (15) using the program SHELXL-93 (16). In the last stage, racemic twinning has been taken into consideration. Additional twinning by a mirror plane, e.g. parallel to (010), or by the equivalent twofold rotation axis, e.g. parallel to [100], was refined.

### **RESULTS AND DISCUSSION**

The refinements converged to a final R1 of 5.64% for 1602 unique reflections ( $I > 2\sigma_I$ ) of the nearly monocrystal specimen 2, data set 2, regarded as a monocrystal, and to 5.59% for its refinement as a rotation twin (relative volume 98.4 and 1.6(2)%), and to 3.15% for 1553 reflections for specimen 1, refined as racemic and rotation twin, data set 1. The crystallographic data and the data of structure refinement of specimen 1 are given in Table 1. A final decision on the real twinning symmetry element of the obtained crystals is not possible.

The structure data obtained for the various refinements only differ within the range of experimental errors. The final results for specimens 1 and 2 are listed in Table 2

TABLE 1
Crystallographic Data and Data of Structure Refinement of
$\gamma$ -In <sub>2</sub> Se <sub>3</sub> (Data Set No. 1)

		-			
	Crystal	data			
Formula weight	466.52	Density	5.449 Mg m <sup>-3</sup>		
Space group	$P6_1$	a =	712.86(5) pm		
Z	6	c =	1938.1(2) pm		
Cell volume	$852.9(1) \times 10^{6} \text{ pm}$	1 <sup>3</sup>			
	Data col	lection			
Temperature	293(2) K	Range of hkl	-10 < h < 8		
Wavelength	71.073 pm	0	0 < k < 10		
Theta range	1.05°-30.04°		-27 < l < 27		
Parameters	49				
	Data of structu	re refinement			
Crystal size (mm	3)	0.13	0.13 imes 0.15 imes 0.15		
Absorption coeff	· · · · · · · · · · · · · · · · · · ·	27.	27.146		
Reflections collec		367	3673		
Independent refle	ections, $R_{int}$	164	1648, 0.050		
Reflections with		155	1553		
F (000)	-	120	1200		
Goodness-of-fit c	on $F^2$	1.0	73		
<i>R</i> 1 ( $I > 2\sigma_I$ ), <i>R</i> 1	(all data)	0.0	0.0315, 0.0350		
$wR2^a$ $(I > 2\sigma_I), v$	$wR2^a$ (all data)	0.0	693, 0.0702		
Extinction coeffic	0.0	0083(8)			
$\Delta \rho_{\min}, \Delta \rho_{\max}$ (10 <sup>-</sup>	<sup>6</sup> e pm <sup>-3</sup> )	-1	.39, 2.61		
	a of the twins (%)	65 :	5:27:3		
-					

 ${}^{a}w = 1/[\sigma^{2}(F_{0}^{2}) + (a * P)^{2} + b * P]$  with a = 0.0301, b = 1.7027, and  $P = [\max(F_{0}^{2}, 0) + 2 * F_{c}^{2}]/3$ .

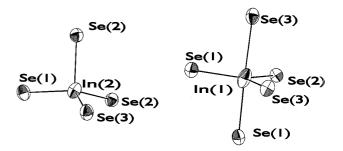


FIG. 1. Coordination polyhedra of the In atoms in  $\gamma$ -In<sub>2</sub>Se<sub>3</sub>. (90% probability).

together with those reported by Likforman *et al.* (8) for comparison; the anisotropic displacement parameters are given in Table 3 and selected bond lengths and angles are given in Table 4.<sup>1</sup> The two  $InSe_n$  polyhedra are shown in Fig. 1 and the arrangement of the polyhedra in the structure are shown in Fig. 2. Three isostructural sulfides, viz. Al<sub>2</sub>S<sub>3</sub> (tbp), AlInS<sub>3</sub>, and GaInS<sub>3</sub>, and isostructural GaInSe<sub>3</sub> have been reported recently (18–21).

As already described in (8), the structure of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> is built up by distorted In(1)Se<sub>5</sub> trigonal bipyramids and In(2)Se<sub>4</sub> tetrahedra. These polyhedra are connected by common corners and edges resulting in a distorted wurtz-ite-type-like arrangement (18). In the crystal structure of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> two different screws are built up by the 6<sub>1</sub> axis. One is formed by corner sharing In(2)Se<sub>4</sub> tetrahedra, and the other by In(1)Se<sub>5</sub> trigonal bipyramids sharing common edges. The screws of one type are interlinked only by those of the other type and vice versa.

The In–Se equatorial distances vary from 257.7 to 262.2 pm, and the axial distances from 287.4 and 297.4 pm (trigonal bipyramid) and 255.2–263.4 pm (tetrahedron), with Se–In–Se angles ranging from 94.55° to 118.88° (tetrahedron). Those of the InSe<sub>5</sub> polyhedron do not largely deviate from 90°, 120°, and 180°, respectively (see Table 4).

The structure proposed for  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> by Likforman *et al.* (8) and, hence, the coordination numbers of the two kinds of In<sup>3+</sup> ions, viz. 4 and 5, are confirmed (see Table 4). This means that because of the higher densities the coordination numbers of both the  $\alpha$ - and  $\beta$ -polymorphs must be greater, i.e., probably 6, than those of the  $\gamma$  modification. These conclusions are in agreement with the results of infrared and Raman spectroscopic measurements, which reveal

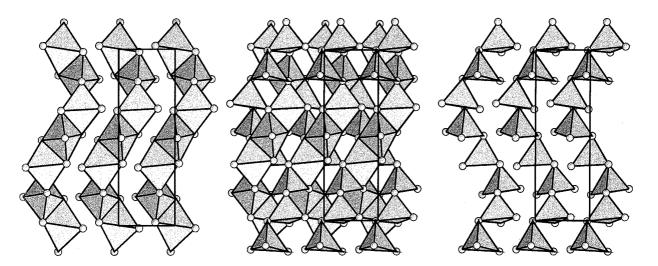
<sup>1</sup> Additional material to this paper can be ordered by referring to CSD 380097, the names of the authors, and citations of the paper at the Fachinformationszentrum Karlsruhe, Gesellschaft für wissenschaftlichtechnische Information mbH, D-76344 Eggenstein-Leopoldshafen, Germany. The list of the  $F_0/F_c$  data is available from the author up to one year after the publication has appeared.

				. –
	x	у	Ζ	$U_{\rm eq} \left[ B_{\rm iso} \right]$
In(1)	0.0117(1)	0.3137(1)	0	2.29(2)
	0.0118(2)	0.3137(2)	0	2.13(3)
	[0.0111(6)	0.3142(7)	0	1.76(2)]
In(2)	0.3417(1)	0.0152(1)	0.30489(5)	1.88(2)
	0.3411(1)	0.0149(2)	0.30476(5)	1.64(3)
	[0.3416(6)	0.0156(7)	0.3055(3)	1.56(2)]
Se(1)	0.0742(2)	0.3470(2)	0.14696(6)	1.78(2)
	0.0739(2)	0.3467(2)	0.14707(7)	1.71(3)
	[0.0728(8)]	0.3470(9)	0.1469(4)	1.33(2)]
Se(2)	0.3473(1)	0.0436(1)	0.1693(1)	1.60(2)
	0.3472(2)	0.0438(2)	0.16950(6)	1.37(3)
	[0.3463(9)	0.0437(8)	0.1703(4)	1.30(2)]
Se(3)	0.3662(2)	0.0250(1)	0.51459(6)	1.93(2)
	0.3656(2)	0.0249(2)	0.51472(8)	1.54(3)
	[0.3646(9)	0.0250(9)	0.5150(3)	1.46(2)]

TABLE 2Atomic Coordinates and Isotropic Displacement Parameters  $U_{eq}$  (10<sup>6</sup> pm<sup>2</sup>)<sup>a</sup>

*Note.* First line, data set No. 1; second line, data set No. 2; data according to Likforman *et al.* (8) in square brackets.

 $^{a}$   $U_{\rm eq}$  is defined as one third of the trace of the orthogonalized  $U_{ij}$  tensor.



**FIG. 2.** Arrangement of the  $InSe_n$  polyhedra in the crystal structure of  $\gamma$ - $In_2Se_3$  (projected along the *b*-axis); screws of the  $In(2)Se_4$  tetrahedra (right) and  $In(1)Se_5$  trigonal bipyramids (left) build up the three-dimensional structure.

Anisotropic Displacement Parameters $U_{ij}$ (100 pm <sup>2</sup> ) <sup>a</sup> of $\gamma$ -In <sub>2</sub> Se <sub>3</sub> (Data Set No. 1)						
	$U_{11}$	$U_{22}$	$U_{33}$	$U_{12}$	$U_{13}$	$U_{23}$
In(1)	1.78(3)	1.51(3)	3.52(4)	0.77(3)	0.72(3)	-0.09(3)
In(2)	1.62(3)	1.75(3)	2.19(3)	0.80(2)	0.33(2)	-0.11(2)
Se(1)	1.57(4)	1.67(4)	2.17(4)	0.88(3)	0.00(3)	0.16(3)
Se(2)	1.57(4)	1.61(4)	1.69(4)	0.84(3)	-0.08(3)	-0.14(3)
Se(3)	1.68(4)	1.59(4)	2.36(4)	0.70(3)	-0.20(4)	0.17(3)

TABLE 3Anisotropic Displacement Parameters  $U_{ij}$  (100 pm<sup>2</sup>)<sup>a</sup> of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> (Data Set No. 1)

<sup>*a*</sup> The anisotropic displacement factor exponent takes the form  $-2\pi^2 \sum_i \sum_j U_{ij} h_i h_j a_i^* a_j^*$ .

TABLE 4Selected Bond Lengths (pm) and Angles (°) of  $\gamma$ -In<sub>2</sub>Se<sub>3</sub>(Data Set No. 1)

$In(1)^{i}-Se(1)^{i}$	287.4(1)	$Se(1)^{i}-In(1)^{i}-Se(1)^{ii}$	92.42(3)
$In(1)^{i}-Se(1)^{ii}$	257.7(1)	$\operatorname{Se}(1)^{i} - \operatorname{In}(1)^{i} - \operatorname{Se}(2)^{iv}$	88.00(4)
$In(1)^{i}-Se(2)^{iv}$	262.2(1)	$Se(1)^{ii}-In(1)^i-Se(2)^{iv}$	121.17(4)
$In(1)^i - Se(3)^{iii}$	258.1(1)	$Se(1)^{i}-In(1)^{i}-Se(3)^{iii}$	90.86(4)
$In(1)^{i}-Se(3)^{v}$	297.4(1)	$Se(1)^{i}-In(1)^{i}-Se(3)^{v}$	172.60(4)
		$Se(1)^{ii}$ -In $(1)^{i}$ -Se $(3)^{iii}$	120.26(4)
		$Se(1)^{ii}$ -In $(1)^{i}$ -Se $(3)^{v}$	88.73(3)
		$Se(2)^{iv}-In(1)^i-Se(3)^{iii}$	118.56(4)
		$Se(2)^{iv}-In(1)^i-Se(3)^v$	85.16(4)
		$Se(3)^{iii}$ - $In(1)^{i}$ - $Se(3)^{v}$	94.88(4)
In(2) <sup>i</sup> -Se(1) <sup>vi</sup>	256.9(1)	$Se(1)^{vi}$ - $In(2)^{i}$ - $Se(2)^{i}$	94.55(4)
$In(2)^{i}-Se(2)^{i}$	263.4(1)	$Se(1)^{vi}$ -In(2) <sup>i</sup> -Se(2) <sup>vii</sup>	115.34(4)
$In(2)^{i}-Se(2)^{vii}$	258.6(1)	$Se(1)^{vi}-In(2)^{i}-Se(3)^{ii}$	118.88(4)
$In(2)^{i}-Se(3)^{ii}$	255.2(1)	$\operatorname{Se}(2)^{i} - \operatorname{In}(2)^{i} - \operatorname{Se}(2)^{vii}$	99.59(3)
		$Se(2)^{i}-In(2)^{i}-Se(3)^{ii}$	111.99(3)
		$Se(2)^{vii}$ - $In(2)^i$ - $Se(3)^{ii}$	112.95(4)

Symmetry transformations used to generate equivalent atoms:

(i) x, y, z; (ii) y, -x + y, z - 1/6; (iii) -x, -y, z - 1/2; (iv) y, -x + y + 1, z - 1/6; (v) -y, x - y, z - 2/3; (vi) x - y + 1, x, z + 1/6; (vii) x - y, x, z + 1/6.

higher frequency lattice modes and, hence, lower coordination numbers of the In<sup>3+</sup> ions in  $\gamma$ -In<sub>2</sub>Se<sub>3</sub> compared to those of the  $\alpha$ ,  $\alpha'$ ,  $\beta$ , and  $\beta'$  polymorphs (5, 17). The results of Osmura *et al.* (7), namely that the indium ions of  $\alpha$ -In<sub>2</sub>Se<sub>3</sub> are located in tetrahedral sites, therefore must be doubted.

## REFERENCES

- S. Popovic, A. Tonejc, B. Grzeta-Plenkovic, B. Celustka, and R. Troiko, J. Appl. Crystallogr. 12, 416 (1979).
- A. Likforman, P. H. Fourcroy, M. Guittard, J. Flahaut, R. Poirier, and N. Szydlo, J. Solid State Chem. 33, 91 (1980).
- C. Julien, M. Eddrief, M. Balkanski, E. Hatzkraniotis, and K. Kambas, *Phys. Status Solidi A* 88, 687 (1985).
- 4. C. Manolikas, J. Solid State Chem. 74, 319 (1985).
- H. D. Lutz, M. Fischer, H.-B. Baldus, and R. Blachnik, J. Less-Common Met. 143, 83 (1988).
- 6. B. Grzeta and S. Popovic, J. Appl. Crystallogr. 23, 340 (1990).
- K. Osamura, Y. Murakami, and Y. Tomiie, J. Phys. Soc. Jpn. 21, 1848 (1966).
- A. Likforman, D. Carré, and R. Hillel, Acta Crystallogr. Sect. B 34, 1 (1978).
- 9. K. Krusch and J. A. Gardner, Phys. Rev. B 24, 4587 (1981).
- 10. C. Julien and M. Eddrief, Mater. Sci. Eng. Sect. B 13, 247 (1992).
- Z. S. Medvedeva and T. N. Guliev, *Izv. Akad. Nauk SSSR, Neorg. Mater.* 1, 848 (1965); *Inorg. Mater.* 1, 779 (1965).
- Ja.Kh. Grinberg, V. A. Boriakova, V. F. Chevel'kov, R. Hillel, and J. Bouix, J. Inorg. Nucl. Chem. 38, 383 (1976).
- 13. M. Fischer, Ph.D. thesis, University of Siegen, Germany (1988).
- E. J. Gabe, Y. LePage, J.-P. Charland, F. L. Lee, and P. S. White, J. Appl. Crystallogr. 22, 384 (1989).
- 15. "International Tables for Crystallography," Vol. C. Kluwer Academic, Dordrecht, 1992.
- G. M. Sheldrick, "SHELXL-93, Program for Crystal Structure Refinement." University of Göttingen, 1993.
- K. Kambas, C. Julien, M. Jouanne, A. Likforman, and M. Guittard, *Phys. Status Solidi B* 124, K105 (1984).
- B. Krebs, A. Schiemann, and M. Läge, Z. Anorg. Allg. Chem. 619, 983 (1993).
- M. Schulte-Kellinghaus and V. Krämer, Acta Crystallogr. Sect. B 35, 3016 (1979).
- 20. P. Dubcek and B. Etlinger, Phys. Status Solidi B 185, 123 (1994).
- G. G. Guseinov, I. R. Amiraslanov, A. S. Kuliev, and K. S. Mamedov, Izv. Akad. Nauk SSSR, Neorg. Mater. 23, 854 (1987).